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Evolution of the structural and mean square displacement parameters in $(NH_4)_x K_{1-x} H_2 PO_4$ solid solutions versus concentration and temperature

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Received 1 October 1997

Abstract. Accurate x-ray data sets were collected at different temperatures for mixed crystals of $(NH4)_x K_{1-x} H_2 PO_4$ (KADP_x), where x = 0.9, 0.8, 0.5, 0.4 and 0.0. The phase transition temperatures of KADP_{0.8} and KADP_{0.9} were observed at $T_c = 109$ K and $T_c = 132$ K respectively. The vibration amplitudes are found to be larger for concentrations corresponding to the glassy phases. These large vibrations result from the static part of the mean square displacement which may be due to a static disorder due to a large difference between cell parameters of pure phases, or to fluctuations of the concentration throughout the crystal. In the range 0.23 < x < 0.75, competition between the two H_2PO_4 configurations occurs (competition between the ferroelectric and antiferroelectric states): this leads to the observed static disorder of H_2PO_4 groups which prevents the phase transition from occurring.

1. Introduction

It is well known that potassium dihydrogen phosphate KH_2PO_4 (KDP) undergoes a parato ferroelectric phase transition at 123 K [1-3], whereas ammonium dihydrogen phosphate $NH_4H_2PO_4$ (ADP) shows a para- to antiferroelectric phase transition at 148 K [4–7]. In the paraelectric phase, both compounds crystallize in the tetragonal system, space group $I\overline{4}2d$. After the transition, they become orthorhombic Fdd2 and $P2_12_12_1$ for KDP and ADP respectively. Mixed ferroelectric and antiferroelectric crystals show a glassy behaviour analogous to a spin-glass system in the intermediate concentrations; $RADP_x$ $(Rb_{1-x}(NH_4)_xH_2PO_4)$ is the most intensively investigated [8] and $(NH_4)_xK_{1-x}H_2PO_4$ $(KADP_x)$ has been established to be analogous to $RADP_x$, but less thoroughly studied. KADP_x shows a ferroelectric phase transition for concentrations close to KDP ($x \leq 0.23$), and antiferroelectric for $x \ge 0.80$ (close to ADP). However, in contrast to RADP_x, the phase diagram is not known accurately [9, 10]. No phase transition was observed in the range 0.23 < x < 0.75, and dielectric measurements on KADP_{0.23} have shown the glassy low-frequency dispersion of dielectric constants [11]. Furthermore, structural studies have established that, for these intermediate concentrations, crystals remain in the paraelectric phase down to 20 K [9,12] in agreement with infrared reflectivity studies [13]. However,

0953-8984/98/071621+21\$19.50 © 1998 IOP Publishing Ltd

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no x-ray or neutron structural analysis in the range $0.23 \le x \le 0.75$ has been carried out at any temperature for KADP_x.

The purpose of this paper is to describe single-crystal x-ray accurate structures of $KADP_x$ and to analyse the atomic mean square displacements versus temperature and concentration, with respect to KDP [15–17] and ADP [17, 18]. Therefore, we report in this paper the results of accurate x-ray diffraction, at three temperatures at least, on six single crystals of $(NH4)_x K_{1-x} H_2PO_4$, corresponding to $x = \{0.0, 0.4, 0.5, 0.8, 0.9, [14], 1.0\}$. This study shows that thermal vibration of the phosphate group varies considerably with composition x, presenting a maximum at x = 0.5, which may be responsible for the glass phase behaviour, and therefore prevents the transition from occurring.

2. Experimental section

Crystals were grown from aqueous solution of ADP and KDP, at fixed temperature (295 K), by slow evaporation. The required ammonium concentration x was determined using the method of Ono *et al* [9]. Because of the non-linear variation of the concentration in aqueous solutions versus solid solutions [9], crystals used for x-ray diffraction were selected at the very beginning of the crystallization process to avoid any fluctuation in the concentration through the crystal. Only crystals that presented an appropriate size, habit and fine diffraction spots on a Weissenberg camera were selected. As expected, the cell parameter c varied with the concentration, which permitted us to check the concentration x for the crystallized materials as shown in figure 1: from this graph, we found that our samples had x very close to the required concentration $x = \{0.0, 0.4, 0.5, 0.8, 0.9, 1.0\}$; all solid solutions present approximately the same habits as ADP and KDP, with $\{100\}, \{010\}, \{101\}$ and $\{011\}$ crystal forms. For x = 0.6 and x = 0.7, no good-quality crystals suitable for accurate x-ray data were obtained, and therefore they were not used in this study.



Figure 1. Variation of cell parameters with ammonium concentration *x*.



Figure 2. Profiles of the (012) reflection against temperature (K): (a) KADP_{0.8}; (b) KADP_{0.9}.

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Table	e 1. Details o	of data colle	ction.				
Formula	$K_{1-x}(NH_4)$	$_{x}H_{2}PO_{4}$					
Space group	I42d						
Diffractometer	CAD4F Enr	af Nonius					
Radiation	$Mo(K\alpha)$ (0.	7107 Å)					
Scan	ω -2 θ						
Ζ	4						
Concentration		x = 0.4			x = 0.5		
Molecular weight (g)		127.66			125.66		
Crystal volume (mm ³)		10.6×10^{-3}	3		13.2×10^{-3}	1	
$\Delta \omega$	1.	15 + 0.35 ta	nθ	1.	2 + 0.35 tar	θ	
$\mu \text{ (mm}^{-1})$		1.16			1.06		
	105 K	158 K	295 K	105 K	158 K	295 K	
a (Å)	7.451(1)	7.458(1)	7.482(1)	7.456(1)	7.463(1)	7.489(1)	
<i>c</i> (Å)	7.145(2)	7.162(2)	7.192(1)	7.214(2)	7.226(2)	7.255(2)	
V (Å ³)	396.7(1)	398.3(1)	402.6(1)	401.0(1)	402.5(1)	406.8(1)	
$(\sin\theta/\lambda)_{max}$ (Å ⁻¹)	1.27	1.27	1.27	1.27	1.27	1.32	
Measured reflections	2002	1999	1977	2038	2429	2274	
Unique reflections							
$(I \ge 3\sigma(I))$	685	632	485	699	645	531	
Scan speed (° mm^{-1})	1.	18 < v < 2.	.75	1.	37 < v < 2.	75	
hkl limit	$0 \leqslant h \leqslant 18$	$0 \leqslant k \leqslant 18$	$l, 0 \leq l \leq 17$	$0 \leqslant h \leqslant 18$	$0 \leqslant k \leqslant 18$	$l \leq l \leq 19$	
Control reflections	(0 8 0), (8	3 0 4), (0 8	4), (8 0 0)	(0 8 2), (8	3 0 4), (0 8	4), (8 0 0)	
Concentration		x = 0.8			<i>x</i> =	= 0.9	
Molecular weight (g)		117.63					
Crystal volume (mm ³)		8.7×10^{-3}			8.8 >	$ (10^{-3}) $	
$\Delta \omega$	1.	.0 + 0.35 tar	θ		1.0 + 0	$.35 \tan \theta$	
$\mu \ (\mathrm{mm}^{-1})$	0.73			0	0.62		
	125 K	158 K	295 K	154 K	164 K	192 K	295 K
a (Å)	7.462(1)	7.470(1)	7.499(1)	7.469(1)	7.471(1)	7.473(2)	7.497(1)
c (Å)	7.409(1)	7.417(1)	7.436(1)	7.456(3)	7.457(3)	7.462(3)	7.471(3)
V (Å ³)	412.5(1)	413.9(1)	418.1(1)	415.9(1)	416.2(1)	416.7(1)	419.9(1)
$(\sin\theta/\lambda)_{max}$ (Å ⁻¹)	1.27	1.27	1.27	1.15	1.22	1.22	1.29
Measured reflections	3209	2065	2196	1552	1828	1875	2290
Unique reflections							
$(I \ge 3\sigma(I))$	638	730	553	575	629	586	492
Scan speed ($^{\circ}$ mm ⁻¹)	0.	86 < v < 2.	75		1.10 <	v < 2.75	
hkl limit	$0 \leqslant h \leqslant 19$	$0 \leq k \leq 19$	$0, 0 \leq l \leq 18$	$0 \leqslant h$	$\leqslant 19, 0 \leqslant 19$	$k \leq 19, 0 \leq l$	≤ 19
Control reflections	(8 0 0), (1	ī 0 0), (0 8	0), (0 8 0)	(4 (0 0), (0 4 0)	, (4 0 6), (5	Ī 6)

Two of these mixed crystals x = 0.9 [14] and 0.8 undergo an antiferroelectric phase transition at 132 and 109 K respectively, as shown by the presence and the growth of the (012) forbidden reflection with temperature (figure 2). The small shift in the scan position is due to the change in cell parameters. The profiles of the forbidden reflections for both concentrations present the same feature: a sudden appearance of the reflection as a function of temperature which is a characteristic of a first-order transition. The transition is reversible as long as the crystal is small enough to avoid existence of domains which lead to destruction of the crystal after the transition. Determination of the high-resolution structure of the antiferroelectric phases is in progress and will be published later.

 Table 2. Internal agreement factors between symmetry equivalent reflections (I, calculated without absorption corrections; II, calculated with absorption correction):

$$R = \left(\sum_{i} |I_i - \langle I \rangle|\right) \left(\sum_{i} I_i\right)^{-1}$$
$$R_w = \left(\sum_{i} w_i (I_i - \langle I \rangle)^2\right) \left(\sum_{i} w_i I_i\right)^{-1}$$
$$w_i = \frac{1}{\sigma(I_i)^2}$$
$$R_2 = R_w \text{ with } w = 1$$

 N_{Term} is the total number of measured reflections, N_{Uniq} is the number of unique reflections, A_{min} and A_{max} are the minimum and the maximum absorption.

								N_{Uniq}	
x	T (K)		R_1 (%)	R_2 (%)	R_w (%)	N _{Term}	N_{Uniq}	$I > 3\sigma(I)$	$A_{min} - A_{max}$
0.4	105	Ι	2.19	2.63	3.17	1894	939	685	0.756-0.790
		II	2.11	2.53	3.07				
	158	Ι	2.33	2.79	3.03	1888	993	632	
		II	2.30	2.69	2.93				
	295	Ι	2.44	2.57	3.05	1870	1002	485	
		II	2.33	2.34	2.97				
0.5	105	Ι	4.23	5.39	4.46	1924	1004	700	0.749-0.814
		II	2.30	2.87	3.62				
	158	Ι	3.65	4.99	3.59	2352	974	645	
		II	2.41	3.64	2.97				
	295	Ι	3.63	4.94	3.51	2100	1116	531	
		II	1.59	2.17	2.03				
0.8	125	Ι	2.21	1.93	2.37	3079	1002	638	0.850-0.869
		II	2.16	1.84	2.35				
	158	Ι	0.96	0.86	1.87	1945	980	730	
		II	0.89	0.66	1.83				
	295	Ι	1.06	1.13	1.76	2079	991	553	
		II	0.88	0.76	1.68				
0.9	154	Ι	1.66	1.84	3.09	1552	756	594	0.847-0.901
		II	1.25	1.20	2.70				
	164	Ι	1.67	1.81	3.09	1828	886	650	
		II	1.32	1.20	2.71				
	192	Ι	2.00	2.23	2.82	1875	887	605	
		II	1.36	1.00	2.47				
	295	Ι	2.37	2.37	2.79	2290	1130	506	
		II	1.33	0.89	2.22				

For each of these concentrations, we measured x-ray diffraction data in the paraelectric phase at three different temperatures at least, using a CAD4F Enraf Nonius [19] diffractometer equipped with a liquid nitrogen vapour stream apparatus installed in a dry box to prevent ice formation. The gas stream temperature was maintained constant within 1 K as monitored by a copper–constantan thermocouple positioned 5 mm upstream from the crystal. The temperature was calibrated using the transition temperature of ADP (148 K) and KDP (123 K). All the data were measured using graphite monochromated Mo(K α) radiation with an $\omega/2\theta$ scan; the width scanned is defined as $\Delta \omega = a + 0.35 \tan(\theta)$, where *a* is a variable depending mainly on the crystal mosaicity and on the divergence of the beam. The values of the *a* parameter which therefore depends on the crystal quality was found to be sensitive to the *x* concentration, larger in glassy phases ($a = 1.15^{\circ}$, 1.20° for x = 0.4, 0.5 respectively) compared to $a = 1^{\circ}$ for the other concentrations studied. This latter value is equal to that used for the pure ADP phase [18]. Details of the data collection are summarized in table 1.

3. Data reduction

Data reduction and error analysis were performed using Blessing's programs DREAR [20], and Gaussian quadrature absorption correction was applied with program Absorb [21]. Data were measured, at least, at a resolution of 1.15 Å⁻¹ (sin θ/λ), and half the Ewald sphere was measured in each case. As shown in table 2, the absorption correction improved well the statistical agreements between equivalent reflections. This correction goes from 1% to 6% and the internal agreement factors are given in table 2. The intensity variances were calculated according to McCandlish *et al* [22]; $\sigma^2(|F|^2) = \sigma_c^2(|F|^2) + p(|F|^2)$ where $\sigma_c^2(|F|^2)$ accounts for Poisson statistics and *p* is an instrumental factor (0.00 < *p* < 0.023).

For x = 0.8, the internal agreement increases from $R_1 = 0.89\%$ (T = 158 K) to 2.16% at T = 125 K. This is probably due to the fact that we are approaching the transition temperature ($T_c = 109$ K); therefore, the crystal starts to present domains that have already transited (domains which should be very small compared to the rest of the crystal). These domains could be microdomains in which the x concentration is slightly higher than 0.8 (non-homogeneity of the crystals).

4. Crystal structure refinements

The structures of $(NH_4)_x K_{(1-x)}H_2PO_4$ (x = 0.4, 0.5, 0.8, 0.9, 1), were determined and refined with SHELX [23] and NRCVAX [24] programs; hydrogen atoms were found by Fourier difference maps. Position and displacement parameters of non-hydrogen atoms were refined at high order $(\sin \theta / \lambda \ge 0.8 \text{ Å}^{-1})$. Hydrogen atoms were refined using low-order data. We found a high correlation between the U^{ij} of the potassium and the ammonium nitrogen atoms, because they occupy the same site (0, 0, 1/2); so, different strategies were adopted in order to determine their thermal parameters and the concentration x: the concentration was determined from the variation of the c parameter (figure 1) and was fixed at the first stage of the refinements; U^{ij} of K and N atoms were refined independently at high angles. After convergence, the N and K occupancies were refined. Another strategy was the use of a mixed form factor, which consists of a weighted sum of the form factors of N and K atoms. This led to U^{ij} values of the average atom statistically equal to the weighted average of those found for N and K separately.

Then, in order to better deconvolute anisotropic mean square displacement parameters from the aspherical atomic electron density, a multipolar analysis of the density was performed, using separate N and K atoms. The multipole refinement consists in expanding the electron density around each centre, according to the Hansen–Coppens model [25]:

$$\rho(\mathbf{r}) = \rho_c(\mathbf{r}) + P_v \kappa^3 \rho_v(\kappa \mathbf{r}) + \sum_{lm} P_{lm} \kappa'^3 R_{nl}(\kappa' \mathbf{r}) y_{lm}(\theta, \phi)$$
(1)

where $R_{nl}(r) = Nr^{nl} e^{-\kappa'\alpha r}$ is a Slater type radial function which has an extension big enough to take into account the bonding electron density, and $y_{lm}(\theta, \phi)$ are real spherical harmonics; $\rho_c(r)$ and $\rho_v(r)$ are spherical core and valence electron distributions, obtained using Clementi [26] wavefunctions for isolated atoms, P_v is the number of valence electrons. The κ and κ' are radial expansion/contraction coefficients of the spherical and aspherical valence deformation densities respectively. By developing the density in this way, we can



(b)

Figure 3. Static deformation electron density maps calculated at 158 K: (a) $KADP_{0.8}$ and (b) ADP. Contour level is $0.05 e^{-} Å^{-3}$.

take into account the fact that electron density around each atom is not spherical because of interactions.

As we had a very good data set for x = 0.8 (see table 2), we have performed a multipolar refinement against these very accurate data at (T = 125, 158 and 295 K). The resulting static model deformation density maps calculated as

$$\Delta \rho(\mathbf{r}) = P_v \kappa^3 \rho_v(\kappa r) - N_v \rho_v(r) + \sum_{lm} \kappa'^3 R_{nl}(\kappa' r) P_{lm} y_{lm}(\theta, \phi)$$
(2)

Table 3. Fractional coordinates and thermal parameters of the atoms for the KADP_{0.8} compound: (a) independent atom model refinements, (b) refinements using multipolar density P_{lm} parameters of KADP_{0.8} at 158 K. Statistics of the refinements: $R(F) = \Sigma \Delta / \Sigma k^{-1} |F_0|$, $R_w(F) = (\chi^2 / \Sigma w F_0^2)^{1/2}$ and $S = (\chi^2 / (N - N_{par}))^{1/2}$, where $\chi^2 = \Sigma w \Delta^2$, $\Delta = |k^{-1}|F_0| - |F_0||$, $w = 1/\sigma^2(F_0)$ and N_0 and N_{par} are respectively the number of data and the number of refined parameters. *x* is the site occupancy and *k* is the scale factor.

		T =	125 K	T = 158 K	T =	295 K
Atom	Parameter	(a)	(b)	(b)	(a)	(b)
N	U^{11}	0.016 02(21)	0.015 96(13)	0.01801(13)	0.03077(22)	0.03021(13)
	U^{33}	0.01297(49)	0.014 01(34)	0.01586(31)	0.023 43(51)	0.025 45(30)
Κ	U^{11}	0.014 33(10)	0.014 42(11)	0.01681(9)	0.029 52(10)	0.029 85(12)
	U^{33}	0.012 20(26)	0.012 02(28)	0.015 17(20)	0.025 16(23)	0.024 21(30)
Р	U^{11}	0.006 12(4)	0.006 20(4)	0.007 84(3)	0.014 04(4)	0.014 06(3)
	U^{33}	0.01974(11)	0.01977(10)	0.02072(7)	0.028 60(8)	0.028 70(7)
0	x	0.147 96(6)	0.147 97(4)	0.147 80(4)	0.14673(6)	0.146 59(3)
	у	0.08431(7)	0.08437(4)	0.08447(4)	0.084 50(6)	0.084 58(3)
	z	0.117 34(10)	0.117 39(5)	0.117 32(6)	0.11731(9)	0.117 14(4)
	U^{11}	0.009 95(12)	0.010 00(9)	0.01202(8)	0.021 25(14)	0.021 30(9)
	U^{22}	0.009 11(12)	0.008 92(9)	0.01091(8)	0.01941(14)	0.019 22(9)
	U^{33}	0.022 22(19)	0.022 44(14)	0.02433(12)	0.037 18(21)	0.03679(11)
	U^{12}	0.000 68(5)	0.000 63(4)	0.00080(3)	0.001 63(6)	0.001 65(4)
	U^{13}	-0.00218(9)	-0.00215(6)	-0.00267(5)	-0.00490(10)	-0.00467(5)
	U^{23}	-0.00144(8)	-0.001 37(6)	-0.00165(5)	-0.00322(9)	-0.00306(5)
H(O)	x	0.150(2)	0.139(2)	0.133(2)	0.141(1)	0.133(1)
	У	0.214(2)	0.225(3)	0.224(2)	0.213(1)	0.224(2)
	z	0.116(4)	0.096(5)	0.100(4)	0.111(2)	0.119(2)
	U	0.023(5)	0.023(5)	0.031(4)	0.028(3)	0.035(3)
H(N)	x	0.014(2)	-0.001(3)	0.001(2)	0.016(1)	0.007(2)
	У	0.085(2)	0.102(2)	0.105(2)	0.088(1)	0.099(1)
	z	0.433(1)	0.411(1)	0.414(1)	0.441(1)	0.418(1)
	U	0.034(5)	0.038(4)	0.043(3)	0.056(2)	0.058(3)
	k	0.8379(5)	0.8414(8)	1.7157(7)	1.7181(6)	1.7211(13)
	R(F) (%)	1.68	1.76	1.32	1.23	1.34
	$R_w(F)$ (%)	1.38	1.48	1.10	0.82	0.97
	G.O.F.	0.78	0.84	1.06	0.96	1.14
	x	0.805(2)	0.808(2)	0.807(1)	0.806(2)	0.808(2)

with N_v = number of valence electrons of the free neutral atom compared to those calculated for ADP [18, 27] did not show any significant differences: the KADP_{0.8}, static deformation map calculated from 158 K data (figure 3) reveals bonding P–O and lone pair density equal to those obtained from the ADP data within standard deviations $(\sigma (\Delta \rho) \approx 0.05 \text{ e}^- \text{ Å}^{-3})$. Therefore we transferred the multipolar density P_{lm} parameters of KADP_{0.8} at 158 K to all other structures (x = 0.4, 0.5 and 0.9) at different temperatures. This transfer consists in assigning an aspherical electron density to P, O, N and H atoms, and by Fourier transformation of (1) we obtain the form factors as follows (Lecomte [25]):

$$f(\boldsymbol{H}) = f_{IAM}(H) + \delta f(\boldsymbol{H})$$

$$\delta f(\boldsymbol{H}) = P_v f_v \left(\frac{H}{\kappa}\right) + \sum_{lm} P_{lm} \Phi_{nl} \left(\frac{H}{\kappa'}\right) Y_{lm\pm} \left(\frac{\boldsymbol{H}}{\boldsymbol{H}}\right)$$
(3)

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. Fractional	8 at 158 K.
Table 4.	KADP _{0.}

			KADP _{0.4}			KADP _{0.5}			KAI	DP _{0.9}	
Atom	Parameter	105 K	158 K	295 K	105 K	158 K	295 K	154 K	164 K	192 K	295 K
z	U ¹¹	0.0118(3)	0.0155(4)	0.0279(7)	0.0114(2)	0.0153(2)	0.0266(6)	0.01665(13)	0.01751(13)	0.01958(14)	0.028 62(21)
	U^{33}	0.0154(7)	0.0191(9)	0.0355(18)	0.0144(4)	0.0198(6)	0.0312(15)	0.013 15(34)	0.013 17(34)	0.01627(37)	0.023 05(53)
К	U^{11}	0.01061(5)	0.01431(6)	0.02551(10)	0.01137(4)	0.01502(5)	0.026 18(12)	0.01648(27)	0.01704(29)	0.01987(30)	0.028 47(50)
	U^{33}	0.01396(10)	0.01720(12)	0.024 52(20)	0.01456(10)	0.01622(13)	0.025 47(30)	0.01454(72)	0.01485(77)	0.01822(84)	$0.024\ 00(143)$
Р	U^{11}	0.00561(4)	0.007 65(4)	0.01323(6)	0.00568(4)	0.00746(4)	$0.013\ 00(10)$	0.007 63(5)	0.00802(5)	0.00935(5)	0.013 64(7)
	U^{33}	0.02709(14)	0.02845(15)	0.03278(19)	0.028 65(13)	0.03040(13)	0.037 07(24)	0.01844(11)	0.01887(12)	0.02060(11)	0.026 14(14)
0	x	0.14851(4)	0.14821(4)	0.14734(6)	0.14842(4)	0.14807(4)	0.14714(7)	0.14782(5)	0.14776(5)	0.14754(4)	0.14676(6)
	У	0.08371(4)	0.08380(4)	0.08377(6)	0.08392(4)	0.08399(4)	$0.084\ 20(8)$	0.08455(5)	0.08451(5)	0.08461(5)	0.084~70(7)
	й	0.12239(8)	0.12224(9)	0.121 69(10)	0.12110(7)	0.12088(7)	0.120 06(12)	0.11625(6)	0.11628(7)	0.11628(6)	0.11615(8)
	U^{11}	0.00815(7)	0.01072(8)	0.01899(14)	0.00840(7)	0.01088(9)	0.018 34(22)	0.011 62(12)	0.01229(11)	0.01384(11)	0.020 56(18)
	U^{22}	0.00784(7)	0.01042(8)	0.01781(13)	0.008 11(7)	0.01037(9)	0.017 08(21)	0.01058(12)	0.01105(11)	0.01269(12)	0.018 87(18)
	U^{33}	0.02710(15)	0.03039(18)	0.03940(25)	0.02883(14)	0.03216(16)	0.04395(34)	0.02204(17)	0.02287(18)	0.02569(17)	0.035 76(28)
	U^{12}	0.00037(3)	0.00062(4)	0.00127(6)	0.00039(3)	0.00067(4)	0.001 27(9)	0.00075(5)	0.00083(5)	0.00087(5)	0.001 58(8)
	U^{13} .	-0.00157(6)	-0.00207(7)	-0.003 72(10) -	-0.00163(6)	-0.00218(6)	-0.003 76(15)	-0.00259(7)	-0.002 84(7)	-0.00318(7)	-0.00489(11)
	U^{23} .	-0.00115(5)	-0.001 59(6)	-0.002 64(10) -	-0.00113(5)	-0.00148(6)	-0.002 60(14)	-0.00156(7)	-0.001 65(7)	-0.00198(7)	$-0.003\ 03(11)$
(O)H	x	0.161(5)	0.161(5)	0.156(4)	a	0.146(3)	а	0.126(4)	0.129(4)	0.124(3)	0.125(4)
	У	0.218(5)	0.218(5)	0.223(4)	a	0.226(3)	а	0.228(3)	0.228(4)	0.226(3)	0.225(4)
	й	0.085(5)	0.089(7)	0.102(8)	а	0.114(5)	а	0.134(4)	0.143(3)	0.139(3)	0.137(4)
	U	0.034(7)	0.038(9)	0.052(8)	0.020(6)	0.034(5)	0.044(5)	0.026(6)	0.029(7)	0.028(6)	0.043(8)
H(N)	x	-0.009(5)	-0.009(8)	-0.005(7)	а	-0.010(6)	а	0.011(3)	-0.011(3)	-0.008(2)	-0.002(4)
	У	0.107(6)	0.094(8)	0.121(8)	a	0.105(6)	а	0.111(3)	0.104(3)	0.106(2)	0.106(3)
	Х	0.422(4)	0.418(5)	0.431(8)	a	0.417(2)	а	0.415(1)	0.414(1)	0.414(1)	0.412(2)
	U	0.044(7)	0.049(10)	0.057(9)	0.057(9)	0.069(7)	0.071(7)	0.049(4)	0.055(5)	0.048(3)	0.054(6)
	k	1.896(1)	1.665(1)	1.896(2)	1.893(2)	2.129(2)	2.161(3)	1.2845(9)	1.2875(9)	1.2946(8)	1.2586(9)
	R(F)	1.82	1.79	1.83	2.17	1.84	2.82	1.81	2.06	2.08	2.29
	$R_w(F)$	1.88	1.92	1.95	2.56	2.07	2.78	1.73	1.83	1.47	1.62
	G.O.F.	1.04	0.94	1.02	0.94	1.06	2.16	1.38	1.33	1.36	1.44
	N_{obs}	685	632	485	669	645	531	575	629	586	492
Conc.	(x)	0.4	0.4	0.4	0.5	0.5	0.5	0.897(2)	0.899(2)	0.899(2)	0.907(3)

^a Same fractional coordinates as KADP_{0.5} at 158 K.



Figure 4. Crystal structure of ADP projected on the *a*, *b* plane (paraelectric phase). Black and grey tetrahedra are NH_4 and PO_4 entities respectively.

where

$$\Phi_{nl}(H) = 4\pi \mathrm{i}^l \int_0^\infty R_{nl}(r) J_l(2\pi H r) r^2 \,\mathrm{d}r.$$

 f_{IAM} is the usual independent atom model form factor. The P_v , κ , κ' and P_{lm} parameters were fixed at the 158 K KADP_{0.8} values and never refined. This strategy led to very precise thermal parameters of the PO₄ group. Tables 3 and 4 give the summary of the refinements for KADP_x (x = 0.8 and x = 0.4, 0.5 and 0.9) respectively. Multipole parameters are available on request to the authors (lecomte@lcm3b.u-nancy.fr).

5. Structural results

In this section, we will discuss our structural results compared to those concerning KDP [17] and ADP [18].

In the paraelectric phase, the KDP structure consists of a network of PO_4 tetrahedra that are linked by very short hydrogen bonds. These hydrogen bonds are almost parallel to the *a* and *b* axes. The potassium is surrounded by six phosphate tetrahedra in a distorded octahedron fashion, two being above and below along the *c* axis. ADP has the same structure



Figure 5. Variation of cell parameters with temperature.

(figure 4) except that the potassium cation is replaced by the ammonium tetrahedron which is therefore linked to phosphate tetrahedra via hydrogen bonds [28]. There is a very small difference between the cell parameters a of ADP and KDP, in contrast to c which varies by about 0.5 Å, because of the large difference between the ionic radius of K⁺ and NH₄⁺ (1.33 and 1.42 Å respectively). So, the a parameter is almost independent of the x concentration, while c varies linearly with concentration x, in agreement with Ono *et al* [9, 11].

5.1. Cell parameters versus temperature

Table 5 gives the variation of the cell parameters with temperature and concentration x. As shown in figure 5 cell parameters decrease linearly with decreasing temperature. This contraction is also dependent on the concentration. The linear thermal expansion, along the a axis defined as

$$\alpha_a = \frac{\mu_a}{a}$$
 with $\mu_a = \frac{\partial a}{\partial T}$

increases very little with the concentration (from 20×10^{-6} to 29×10^{-6} K⁻¹) (table 6); then the rate of change is hardly dependent on the concentration as expected.

Table 5. Cell parameters against concentration and temperature.

Tomporatura	<i>x</i> =	0.4	<i>x</i> =	0.5	<i>x</i> =	0.8	<i>x</i> =	= 0.9
(K)	a (Å)	c (Å)						
105	7.451(1)	7.145(2)	7.456(1)	7.214(2)				
110	7.452(1)	7.151(2)	7.457(1)	7.215(2)				
125					7.462(1)	7.409(1)		
130	7.454(1)	7.155(2)	7.459(1)	7.219(2)				
147	7.456(1)	7.159(2)	7.460(1)	7.223(2)				
154	. ,	. ,	. ,	. ,			7.469(1)	7.456(3)
158	7.458(1)	7.162(2)	7.463(1)	7.226(2)	7.470(1)	7.417(1)		
164							7.471(1)	7.457(3)
170			7.465(1)	7.228(2)			. ,	
180	7.460(1)	7.166(2)	7.466(1)	7.230(2)	7.474(1)	7.421(1)		
192	. ,	. ,	. ,	. ,	. ,	. ,	7.473(2)	7.462(3)
200			7.470(1)	7.236(1)				
220	7.467(1)	7.176(1)	7.473(1)	7.241(1)	7.481(1)	7.427(1)		
234					7.484(1)	7.430(1)		
295	7.482(1)	7.192(1)	7.489(1)	7.255(1)	7.499(1)	7.436(1)	7.497(1)	7.471(3)

Table 6. Thermal expansion/contraction against ammonium concentration x.

Parameter	Coefficient	x = 0	x = 0.4	x = 0.5	x = 0.8	x = 0.9	x = 1.0
a	$\mu_a (10^{-4}) \text{ Å } \text{K}^{-1}$	1.51	1.60	1.73	2.06	1.90	2.21
с	$\alpha_a (10^{-4}) \text{ K}^{-1}$ $\mu_c (10^{-4}) \text{ Å K}^{-1}$	20 2.53	21 2.36	23 2.19	26 1.57	25 0.80	29 0.52
	$\alpha_c \ (10^{-6}) \ \mathrm{K}^{-1}$	36	33	30	21	11	6.6

Along the *c* axis, as a function of *x*, α_c decreases non-linearly by a large amount from $36 \times 10^{-6} \text{ K}^{-1}$ to $7 \times 10^{-6} \text{ K}^{-1}$, i.e. 500% (table 6). For a high ammonium concentration, the change in *c* with temperature is small, and for low concentration the change is larger. This is due to the type of interaction between ammonium or potassium cations with the PO₄ anions. For KDP, the interaction is ionic, and therefore the K...PO₄ distance is more temperature dependant than the NH₄...PO₄ distance in ADP where the interaction is a combination of ionic and hydrogen bond interactions: because of the hydrogen bonds, the rate of change is small; in addition, the ammonium group has a larger ionic radius than that of the potassium cation.

5.2. Variation of the stereochemical parameters with temperature

Table 7 shows the evolution of the main distances and angles as a function of temperature and concentration. For the sake of comparison, table 8 shows the values of these distances and angles calculated for ADP and KDP. At first, we see that the bond distances and angles show similar behaviour as a function of T whatever the concentration: the distance P–O does not change within standard deviations as already stated by Nelmes *et al* [15, 16] for KDP in a temperature and pressure dependent study. The O^a ... Oⁱ distance (short hydrogen bond) (see tables 7 and 8 for nomenclature) increases with the temperature by 0.01 Å (about 10σ), and the θ angle (figure 6) varies by 0.25 degree. The N ... O distances which are involved in hydrogen bonds with adjacent PO₄ groups increase with temperature by 0.016, 0.019, 0.019 and 0.018 Å, for x = 0.4, 0.5, 0.8 and 0.9 respectively, when the temperature varies from 158 to 295 K. These variations are similar to those observed in pure ADP (0.020 Å) [18].



Figure 6. Projections of PO₄ group in (001) plane (see table 7 for nomenclature).

5.3. Variation of the stereochemical parameters with concentration

When x goes from 0 to 1, the main variations are the following: the O^a-P-O^b angle increases by 0.8° and O^a-P-O^c decreases by 0.4° ($\sigma = 0.02^\circ$); the N(K)... $O^{a,b,c,d}$ and N(K)... $O^{e,f,g,h}$ (see tables 7 and 8) distances increase whatever the temperature; for example, at T = 158 K these distances are equal to 2.8424(5) and 2.9821(6) Å for x = 0.4 compared to 2.8896(2) and 3.1664(2) Å in ADP [18]. Then, N... $O^{e,f,g,h}$ varies by more than 6%; the P–O...N(K) angle also increases from 98.70(1)° in KDP to 100.62(1)° for ADP (1.9%) while P...N(K)...O decreases by almost 8% (25.70° to 23.70(1)°).

Then, the increase of the *c* parameter with ammonium concentration results from the variation of the N–O interactions as well as from the variation of P–O...N(K) and P...N(K)...O angles. All these observations are in fact related to the bifurcated P–O...H–N hydrogen bonds which could play an important role in the mechanism of the phase transition [28].

		KADP _{0.4}			KADP _{0.5}			KADP _{0.8}			KAI	0P _{0.9}	
	105 K	158 K	295 K	105 K	158 K	295 K	125 K	158 K	295 K	154 K	164 K	192 K	295 K
(N)H–N	0.94(1)	0.91(5)	1.03(3)		0.99(4)		1.01(1)	1.01(1)	0.97(1)	1.04(2)	1.02(2)	1.02(1)	1.03(2)
0-H(0)	1.04(3)	1.03(4)	1.05(3)		1.06(2)		1.06(2)	1.06(2)	1.06(1)	1.09(3)	1.10(3)	1.09(2)	1.08(3)
0-0i	2.4784(5)	2.4794(7)	2.4881(10)	2.4771(6)	2.4785(7)	2.4847(13)	2.4745(6)	2.4757(6)	2.4837(5)	2.4750(8)	2.4761(8)	2.4753(8)	2.4820(11)
(O)H-(O)H	0.75(8)	0.70(9)	0.52(8)		0.39(5)		0.57(6)	0.54(5)	0.39(2)	0.35(5)	0.41(6)	0.40(4)	0.41(6)
P-Oa	1.5421(4)	1.5423(4)	1.5411(6)	1.5417(4)	1.5417(4)	1.5418(7)	1.5401(3)	1.5409(4)	1.5392(3)	1.5392(4)	1.5391(4)	1.5389(4)	1.5384(6)
$N-O^{e,f,g,h}$	2.8424(5)	2.8481(5)	2.8648(6)	2.8569(5)	2.8569(5)	2.8756(7)	2.8744(4)	2.8791(3)	2.8985(3)	2.8831(5)	2.8839(5)	2.8868(5)	2.9012(6)
$N-O^{a,b,c,d}$	2.9821(6)	2.9886(6)	3.0020(8)	3.0197(6)	3.0197(6)	3.0406(10)	3.1065(4)	3.1102(5)	3.1170(4)	3.1312(6)	3.1311(6)	3.1327(5)	3.1365(7)
H(N)–O ^f	1.94(5)	2.03(6)	1.88(3)	1.95(4)	1.95(4)	1.97(4)	1.99(2)	1.99(1)	2.06(1)	1.92(2)	1.96(3)	1.96(2)	1.99(3)
O-P-Ob	108 76(2)	108 80(2)	108 82(3)	108 72(2)	108 72(2)	108 77(4)	108 59(2)	108 59(2)	108 68(1)	108 49(2)	108 50(2)	108 54(2)	108 55(3)
O_P_O	11091(3)	110 83(3)	110 79(4)	110.98(3)	110.98(3)	110 89(5)	(-)	(2)	(111 07(2))	11146(3)	111 43(3)	11135(2)	111 33(3)
P-0-N	99.34(2)	99.45(2)	99.61(3)	99.57(2)	99.63(2)	99.87(3)	100.23(2)	100.25(1)	100.44(1)	100.31(2)	100.32(2)	100.39(2)	100.44(3)
P-N-O	25.21(1)	25.14(1)	24.99(1)	24.91(1)	24.88(1)	24.68(1)	24.15(1)	24.14(1)	24.03(1)	23.97(1)	23.96(1)	23.94(1)	23.89(1)
P-O-H(O)	108(2)	109(3)	112(2)	112(2)	112(2)	112(2)	106(2)	116(1)	110(1)	111(1)	114(1)	112(1)	112(2)
O-(N)H-N	153(3)	147(4)	158(4)		151(2)		145(1)	146(1)	144(1)	152(1)	150(1)	149(1)	147(2)
0-H(0)-0 ¹	151(4)	154(5)	163(5)		172(3)		159(3)	161(2)	170(1)	164(2)	161(3)	161(1)	162(3)
H(N) - N - M(N)	108(3)	114(5)	103(3)		111(3)		115(2)	114(1)	114(1)	112(1)	113(2)	113(1)	114(2)
$H(N)-N-H(N)^{c}$	110(3)	100(4)	122(3)		106(2)		99(1)	102(1)	101(1)	105(1)	102(1)	102(1)	101(2)
θ	60.59(1)	60.52(1)	60.39(2)		60.43(1)		60.30(1)	60.25(1)	60.02(1)	60.23(2)	60.24(1)	60.17(2)	60.01(2)

Table 7. Bond distances (\hat{A}), angles ($^{\circ}$) and standard deviations in KADP_x (x = 0.4, 0.5, 0.8 and 0.9): (a) x, y, z; (b) y, \bar{x}, \bar{z} ; (c) \bar{x}, \bar{y}, z ; (d) $\bar{y}, x, 1 - z$; (e) 1/2 - x, y, 3/4 - z;

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	4 DP [18 ^b]	ADP [20]	ADP [28]	KDF	[17]
	158 K	152 K	neutron 158 K	127 K	295 K
N–H(N)	0.99(1)	0.94(1)	1.002(5)		
O-H(O)	1.07(1)	1.13(2)	1.049(3)	1.0712(7)	1.0677(16)
O–O ⁱ	2.4776(4)	2.4858(9)	2.471(4)	2.4829(4)	2.4946(6)
H(O)-H(O)	0.33(3)	0.26(4)	0.377(5)	0.3428(12)	0.3647(28)
P–O ^a	1.5399(2)	1.5395(5)	1.538(2)	1.5429(3)	1.5403(4)
N–O ^{e,f,g,h}	2.8896(2)	2.9004(4)	2.889(3)		
N–O ^{a,b,c,d}	3.1664(3)	3.1298(6)	3.172(4)		
H(N)-O ^f	1.95(1)	2.01(2)	1.943(6)		
O-P-O ^b	108.52(1)	108.80(2)	108.57(4)	108.92(1)	108.95(1)
O-P-O ^c	111.39(1)	110.8(3)	111.29(7)	110.58(3)	110.52(3)
P-O-N(K)	100.62(1)		100.76(4)	98.57(1)	98.86(1)
P–N(K)–O	23.70(1)		23.60(4)	25.62(1)	25.88(1)
P-O-H(O)	115.6(6)	121(1)	116.3(2)	112.34(6)	112(1)
N–H(N)–O	157.7(6)		156.4(4)		
O-H(O)-O ⁱ	178(1)	169(2)	176.5(5)	178.22(16)	177.23(40)
H(N)-N-H(N) ^b	108(1)	107(1)	109.2(5)		
H(N)-N-H(N) ^c	112.3(6)	115(1)	109.6(2)		
θ	60.03(1)	60.04(1)	60.16	60.98(1)	60.89(1)

Table 8. Bond distances, angles and standard deviations for ADP and KDP.

6. Mean square displacement parameters: analysis and discussion

The most important feature in these compounds is the variation of mean square displacement amplitudes with ammonium concentration: for all temperatures, the anisotropic thermal vibration along the c axis present a maximum for x close to 0.5.

6.1. Variation of U^{ii} with temperature

 U^{11} , U^{22} and U^{33} for all P, O, N(K) atoms vary linearly with temperature (figure 7), showing quasiharmonic mean square displacements for all concentrations studied. For harmonic or nearly harmonic mean square displacements, each U parameter should be proportional to the temperature in the high-temperature limit [15]; then, one can write U as

$$U(T) = U_{stat} + U_{dyn}(T)$$

where U_{stat} is the extrapolation of the mean square displacement at T = 0 K: it represents the static disorder of the atom at T = 0 K. $U_{dyn}(T)$ is proportional to T, and can be related to the dynamical behaviour of the vibration that almost vanishes at T = 0 K. Then if, for example, an atom is moving along the c axis in a double-minimum potential well with minima Δ apart, we expect U_{stat} to be equal to $(\Delta/2)^2$. By inspection of U^{33} , Nelmes *et al* [15, 16] showed that for KDP, P and O atoms occupy two different sites along the c axis for which the largest distance is $\Delta = 0.1$ Å for the P atom. On the other hand, the extrapolation of U^{11} (U^{22}) at zero temperature was found very small, assigned to the zero-point vibration [15, 16].

Our results on mixed crystals agree quite well with those of Nelmes *et al* [15, 16]. For the P and O atoms, the extrapolations of U^{11} (U^{22}), at T = 0 K, are small and independent of concentration (table 9); along the *c* axis, the variation of U^{33} with concentration is surprisingly high: $\Delta(P)$ varies from 0.11 (x = 1) to 0.31 Å (x = 0.5) and Δ (O) goes



(b)

Figure 7. Variation of (a) $U^{11}(P)$ and (b) $U^{33}(P)$ with temperature (x = 0, 0.4, 0.5, 0.8, 0.9, 1).

from 0.09 to 0.28 Å. This extrapolation at T = 0 K, which leads to a large value of U^{33} , demonstrates that the static disorder of the PO₄ group increases with concentration. These results agree with those obtained from neutron inelastic scattering [30–32] which showed that the PO₄ tetrahedron has a large vibration along *c*. This disorder does not show up clearly on the diffraction pattern; no satellite peaks and no significant diffuse scattering appear.

Table 9. Δ_{11} and Δ_{33} distances (Å) between the two atomic sites of P, O against *x* concentration along *a* and *c* directions.



Figure 8. Variation of $U^{ii}(O)$, $U^{ii}(P)$ with concentration at room temperature and at T = 158 K.

6.2. Variation of U^{ii} with concentration

It is interesting to analyse mean square displacements against x at fixed temperature. Figure 8 gives the variation of U^{ii} for the P and O atoms at room temperature and at T = 158 K (the KDP values are extracted from the papers of Nelmes *et al* [15, 17]). The behaviour of U^{11} , U^{22} of P and O is almost linear with x whereas we observe a negative parabolic shape in the U^{33} case, with a maximum for x = 0.5. For K or N, U^{ii} parameters show the same feature with slightly less accuracy.

We can also write U^{33} as the sum of the static and dynamic components of the mean



Figure 9. (a) Static and (b) dynamic parts of $U^{33}(P)$ and $U^{33}(O)$ against x at T = 158 K.

square anisotropic displacement parameters along c:

$$U^{33} = U^{33}_{stat} + U^{33}_{dyn}.$$

The static U_{stat}^{33} and dynamic U_{dyn}^{33} components against concentration are shown in figure 9 (T = 158 K). The static one has a parabolic shape, while the dynamic is mostly linear as x goes from 0 to 1. Then, the non-linear behaviour is attributed to the static part only. Furthermore, when the ratio $U_{stat}^{33}/U_{dyn}^{33}$ for both O and P is close to two, the glassy phase appears. In conclusion, we have shown that, in the paraelectric phase of $K_{1-x}(NH_4)_xH_2PO_4$, the differences in bond lengths, intermolecular interactions and angles as a function of T or x are not contrasted enough within the phase diagram to understand



Figure 10. KH_2PO_4 and $(NH_4)H_2PO_4$ units: (a) paraelectric phases, (b) ferro- (antiferro-) electric phases.

the appearance of the glassy phase. Only a careful study of the thermal parameters allows some conclusions. When the static part of U^{33} of the atoms is predominant, the disorder to order transition cannot occur and a glassy phase appears. In that sense, U^{33} may be considered as the order parameter. Hydrogen bonds play an important role in the phase transition: figure 10 gives the structural arrangement of KDP and ADP in their paraelectric and ferro(antiferro)electric phases. In KADP_x and for x greater than 0.8, the $(NH_4)^+$ ion imposes the H₂PO₄ configuration when the phase transition occurs (as in ADP): the four hydrogen bonds between the ammonium and the neighbouring PO₄ groups in the paraelectric phase (figures 4, bottom, and 11) are reduced to two after the transition imposing the $H_2PO_4^$ configuration. In KDP and for solid solutions with x < 0.23 the K⁺ interacts with H₂PO₄⁻ by coulombic forces. When the transition occurs, in contrast to ADP, the K⁺ ion (0, 0, 1/2) does not impose the configuration of the H₂PO₄⁻ groups. The configuration adopted is the most stable one of the $H_2PO_4^-$ ion itself [33]. In the intermediate region of the phase diagram competition between the ferro- and antiferroelectric states occurs (i.e. between the two H_2PO_4 configurations); therefore, crystallographically, this frustration leads to the observed static disorder of the H_2PO_4 group which prevents the phase transition from occurring and which is revealed by the behaviour of U^{33} .



Figure 11. Projections of PO₄ and (NH₄) groups in the (001) plane. (a) 1/2 + x, \bar{y} , 3/4 - z; (b) -1/2 + x, \bar{y} , 3/4 - z; (c) \bar{x} , 1/2 + y, 1/4 - z; (d) \bar{x} , -1/2 + y, 1/4 - z.

Further work to explore the boundaries of the antiferro–glass and ferroelectric–glass regions by x-ray, neutron and IR techniques is under way.

Acknowledgments

We are grateful to CNRS and CNR for a joint cooperative grant (No 4552) between France and Morocco. A Boukhris is grateful to University Henri Poincaré for an invited Maître de Conférences position.

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